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Traction force microscopy on soft elastic substrates: A guide to recent computational advances☆



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ABSTRACT

The measurement of cellular traction forces on soft elastic substrates has become a standard tool for many labs working on mechanobiology. Here we review the basic principles and different variants of this approach. In general, the extraction of the substrate displacement field from image data and the reconstruction procedure for the forces are closely linked to each other and limited by the presence of experimental noise. We discuss different strategies to reconstruct cellular forces as they follow from the foundations of elasticity theory, including two-versus three-dimensional, inverse versus direct and linear versus non-linear approaches. We also discuss how biophysical models can improve force reconstruction and comment on practical issues like substrate preparation, image processing and the availability of software for traction force microscopy. This article is part of a Special Issue entitled: Mechanobiology.

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1. Introduction

Over the last two decades, it has become apparent that mechanical forces play a central role for cellular decision-making, leading to the emerging field of mechanobiology [1,2]. In order to understand how forces impact cellular processes, it is essential to measure them with high spatiotemporal resolution and to correlate them either statistically or causally with the cellular process of interest. The most common approach is to measure forces at the cell-matrix interface. This field has grown rapidly over the last years and has become to be known as traction force microscopy (TFM). Using this approach, it has been shown e.g. that cellular traction often correlates with the size of adhesion contacts [3–7] but also that this correlation depends on the growth history of the adhesion contact under consideration [8,9]. For most tissue cell types, high extracellular stiffness correlates with large traction forces and large cell-matrix adhesion contacts. These large contacts are thought to not only ensure higher mechanical stability, but also to reflect increased signaling activity. This leads to a stiffness-sensitive response of cells, e.g. during cell spreading and migration [10,11] or stem cell differentiation [12–15]. While TFM has become a standard tool in many labs working on mechanobiology, in practice the details of its implementation vary significantly and the development of new approaches is moving forward at a very fast pace.

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From a general point of view, forces are not an experimentally directly accessible quantity but have to be infered from the fact that they create some kind of motion. Despite the fact that this motion can follow different laws depending on the details of the system under consideration (e.g. being elastic or viscous), a force measurement essentially requires to monitor some kind of dynamics. This is illustrated best with a linear elastic spring. Here force is defined as F = kx, with spring constant *k* and displacement *x*. Without a measurement of *x*, no statement on *F* would be possible (*k* is a constant that can be obtained from a calibration experiment). In order to measure *x*, the reference state x = 0 has to be known, and therefore one typically needs a relaxation process to determine the absolute value of x. Thus even seemingly static situations require some dynamical measurement. Another instructive example is the stress acting over a fictitious surface inside a static but strained elastic body. In order to measure this stress directly, in principle one has to cut the surface open and to introduce a strain gauge that measures forces by the movement of a calibrated spring. Alternatively one needs to use a model that allows one to predict this stress from an elastic calculation.

In summary, each direct measurement of cellular forces has to start with the identification of a suitable strain gauge. Thus a helpful classification of the wide field of TFM can be introduced by considering the different ways in which a strain gauge can be incorporated in a cell culture setup (Fig. 1). The most obvious way to do this is to replace the glass or plastic dishes of cell culture by an elastic system that can deform under cell forces. Early attempts to do so used thin elastic sheets, which buckle under cellular traction and thus provide an immediate visual readout

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Fig. 1. Schematic representation of different setups for traction force microscopy. (a) Thin films buckle under cell traction, therefore this setup is difficult to evaluate quantitatively. (b) The standard setup are thick films with embedded marker beads as reviewed here. The substrate deformation field can be extracted with image processing and has to be deconvoluted to obtain the cellular traction field. (c) Pillar arrays are local strain gauges and do not require any deconvolution; however, they also present topographical and biochemical patterns to cells. (d) Fluorescent stress sensors typically use the relative movement of two molecular domains connected by a calibrated elastic linker to create a fluorescent signal, e.g. by Förster resonance energy transfer (FRET) or by quenching.

(Fig. 1 a) [16]. However, due to this non-linear response, it is difficult to evaluate these experiments quantitatively. Therefore this assay was first improved by using thin silicone films under tension [17] and then thick polyacrylamide (PAA) films that do not buckle but deform smoothly under cell traction (Fig. 1 b) [18]. Today the use of thick films made of different materials is a standard approach in many mechanobiology labs. Fiducial markers can be embedded into these substrates and their movement can be recorded to extract a displacement field. Solving the inverse problem of elasticity theory, cellular traction forces can be calculated from these data [18–22]. An interesting alternative to solving the inverse problem is the direct method that constructs the stress tensor by a direct mapping from a strain tensor calculated from the image data [23–25]. Here we will review these methods that are based on the experimental setup shown in Fig. 1 b.

A simple alternative to TFM on soft elastic substrates is the use of pillar arrays, where forces are decoupled in an array of local strain gauges (Fig. 1 c) [4,7,26–28]. Pillars can be microfabricated from many different materials, including elastomeres like polydimethylsiloxane (PDMS) or solid material like silicium, as long as they have a sufficiently high aspect ratio to deform under cellular traction. One disadvantage of this approach is that cells are presented with topographical cues and that their adhesion sites grow on laterally restricted islands, making this system fundamentally different from unconstrained adhesion on flat substrates. Moreover it has recently been pointed out that substrate warping might occur if the base is made from the same elastic material, thus care has to be taken to correctly calibrate these systems [29].

A very promising alternative to macroscopically large elastic strain gauges is the use of molecular force sensors (Fig. 1 d) [30–36]. Such a sensor typically consists of two molecular domains connected by a calibrated elastic linker. In the example for an extracellular sensor shown in Fig. 1 d, the distal domain is bound to a gold dot on the substrate that quenches the cell-bound domain and fluorescence ensues as the linker is stretched by cellular forces [35]. For intracellular sensors, one can use Förster resonance energy transfer (FRET), which means that fluorescence decreases as the linker is stretched [30]. Fluorescent stress sensors give a direct readout of molecular forces, but for several reasons one has to be careful when interpreting these signals. First the effective spring constant of the elastic linker might depend on the local environment in the cell, even if calibrated in a single-molecule force spectroscopy experiment. Second the fluorescent signal is a sensitive function of domain separation and relative orientation, thus a direct conversion into force can be problematic. Third it is difficult to control the number of engaged sensors, thus the fluorescent signal cannot easily be integrated over a larger region. Fourth the molecular stress sensor reads out only part of the force at work in the cellular structure of interest (e.g. the adhesion contact). Therefore fluorescent stress sensors are expected to complement but not to replace traditional TFM in the future.

One advantage of fluorescent stress sensors over soft elastic substrates and pillar assays is that they can be more easily adapted to force measurements in tissue, for example in developmental systems with fast and complicated cell rearrangements, although the same issues might apply as discussed above for single cells. Recently, however, it has been shown that macroscopic oil droplets can be used to monitor forces during developmental processes [37]. In principle, also subcellular structures such as focal adhesions, stress fibers, mitochondria or nuclei can be used as fiducial markers for cell and tissue deformations [38,39]. One disadvantage of this approach however is that subcellular structures are usually highly dynamic and can exhibit their own modes of movement, thus not necessarily following the overall deformation of the cell. Nevertheless conceptually and methodologically these approaches are similar to traditional TFM and also work in the tissue context.

Another important subfield of TFM is estimating internal forces from cell traction using the concept of force balance. This concept has been implemented both for forces between few cells [40,41] and for forces within laterally extended cell monolayers [42,43]. In the latter case (monolayer stress microscopy), one assumes that the cell monolayer behaves like a thin elastic film coupled to the underlying matrix by stresses (alternatively one can assume coupling by strain [44,45]). Combined with a negative pressure that represents the effect of actomyosin contractility, the physics of thin elastic films is now increasingly used to describe forces of cell monolayers in general [46–49]. Recently single cell and monolayer approaches for internal force reconstruction have been combined by tracking each cell inside a monolayer [50]. For single cells, the combination of modeling and TFM has recently been advanced to estimate the tensions in the whole set of stress fibers within cells on pillar arrays [51] and soft elastic substrates [52]. For the latter case an actively contracting cable network constructed from image data has been employed to model contractility in the set of stress fibers within U2OS cells.

Despite the many exciting developments in the large field of TFM, the most commonly used setup to measure cellular forces is traction force microscopy on soft elastic substrates [53–55]. Here we review the underlying principles and recent advances with a special focus on computational aspects. For the following, it is helpful to classify the different approaches in this field. Because cells become rather flat in mature adhesion on elastic substrates, traditionally only tangential deformations have been considered (2D TFM). More recently, tracking of bead movements in all three dimensions has been used to reconstruct also z-direction forces (3D TFM) [23,24,56-58]. For both 2D and 3D TFM, one further has to differ between linear and non-linear procedures. The central quantity in this context is strain ε , which is defined as relative deformation and therefore dimensionless. If the substrate is sufficiently stiff or the cell sufficiently weak to result in strain values much smaller than unity ($\varepsilon \ll 1$), one is in the linear regime and can work with the small strain approximation [18-22]. The standard approach to estimating forces is the solution of the inverse problem of linear elasticity theory (inverse TFM). In the linear case, one can use the Green's function formalism that leads to very fast and efficient algorithms for force reconstruction using inverse procedures, both for the standard case of thick substrates and for substrates of finite thickness (GF-TFM) [59-61]. If in contrast one is in the regime of large strain ($\varepsilon \gg 1$), Green's functions cannot be used. One way to deal with this problem is the use of the finite element method (FEM-TFM) in a non-linear formulation. Moreover FEM-TFM can also be used in a linear formulation [62,63]. For both small and large strains, alternatively the stress can in principle be constructed directly from the displacement (direct TFM) [23-25,64].

In all approaches used, an important issue is the role of noise on the force reconstruction. Most TFM-approaches use an inversion of the elastic problem to calculate forces from displacement. However, this is an ill-posed problem in the sense that due to the long-ranged nature of elasticity, the calculated traction patterns are highly sensitive to small variations in the measured displacement field and thus solutions might be ambiguous in the presence of noise [20]. Also nonconforming discretization of the problem can cause ill-posedness, e.g. if the mesh is chosen too fine compared to the mean distance of measured bead displacements. In order to avoid ambiguous solutions, a regularization procedure has to be employed in one way or the other, e.g. by filtering the image data or by adding additional constraints to the force estimation [21]. In model-based TFM (MB-TFM), this problem can be avoided if the model is sufficiently limiting [52]. In direct TFM, care has to be taken how to calculate the derivatives from noisy data [25]. Traction reconstruction with point forces (TRPF) can be considered to be a variant of MB-TFM, but requires regularization if one uses many point forces [20,21].

In Table 1 we list the different variants of TFM on soft elastic substrates and their respective abbreviations. In the following we will review recent advances in this field along the lines of this classification. We start with an introduction into TFM on soft elastic substrates, including an explanation of inverse TFM versus direct TFM. We next discuss GF-TFM, FEM-TFM and MB-TFM in more detail. We then discuss some

practical issues arising for TFM-users, namely experimental protocols, image processing and software for force reconstruction. We close with discussion and outlook.

2. Basic principles of elasticity theory and TFM

The reconstruction of traction forces on soft elastic substrates requires continuum elasticity theory to describe substrate deformations as a consequence of forces being applied to the substrate surface [65, 66]. Forces applied to the boundary of elastic solids are called traction forces **t** and are measured in units of force per area, $Pa = N/m^2$. They lead to substrate deformations which in general are described by the deformation gradient tensor **F**, which is the Jacobian of the coordinate transformation from the undeformed state **x** to the deformed state **x**':

$$\mathbf{F} = \begin{pmatrix} \frac{\partial x'}{\partial x} & \frac{\partial x'}{\partial y} & \frac{\partial x'}{\partial z} \\ \frac{\partial y'}{\partial x} & \frac{\partial y'}{\partial y} & \frac{\partial y'}{\partial z} \\ \frac{\partial z'}{\partial x} & \frac{\partial z'}{\partial y} & \frac{\partial z'}{\partial z} \end{pmatrix}.$$
(1)

Alternatively one can work with the displacement vector field $\mathbf{u} = \mathbf{x}' - \mathbf{x}$:

$$\mathbf{F} = \mathbf{1} + (\nabla \otimes \mathbf{u})^{I}. \tag{2}$$

Here **1** is the unit tensor, \otimes is the dyadic product that maps two vectors into a tensor ($(\mathbf{a} \otimes \mathbf{b})_{ij} = a_i b_j$) and ∇ the gradient operator in the undeformed (Lagrangian) frame.

Both **F** and **u** are defined at any point in the body and therefore are spatial fields. They have to be constructed from the image data and then can be used to derive measures for local changes in distances and angles during deformation. For this purpose it is useful to define several types of non-linear strain tensors, in particular the Green–Lagrange tensor **E** and the left Cauchy–Green tensor **B**:

$$\mathbf{E} = \frac{1}{2} \left(\mathbf{F}^{T} \cdot \mathbf{F} - \mathbf{1} \right), \quad \mathbf{B} = \mathbf{F} \mathbf{F}^{T}.$$
(3)

In terms of the displacement vector field, the Green–Lagrange tensor can be written as

$$\mathbf{E} = \frac{1}{2} \left((\nabla \otimes \mathbf{u}) + (\nabla \otimes \mathbf{u})^T + (\nabla \otimes \mathbf{u}) (\nabla \otimes \mathbf{u})^T \right)$$
(4)

which again shows its non-linear nature. Linearization for small strains gives the linear strain tensor:

$$\boldsymbol{\varepsilon} = \frac{1}{2} \left(\left(\nabla \otimes \mathbf{u} \right) + \left(\nabla \otimes \mathbf{u} \right)^T \right).$$
(5)

Table 1

Abbreviations for different variants of traction force microscopy (TFM) on flat elastic substrates as discussed in this review. The corresponding references are given in the main text. Regularization is required to deal with experimental noise. The first five entries in the list exist in both linear and non-linear versions. The other entries are typically used with a linear substrate model.

Abbreviation	Meaning	Comment
2D TFM	Two-dimensional TFM	Standard approach using only tangential deformations and forces
3D TFM	Three-dimensional TFM	Tracking of 3D deformations allows one to reconstruct 3D forces
iTFM	Inverse TFM	Standard approach solving the ill-posed inverse problem of elasticity
dTFM	Direct TFM	If superior 3D image quality is available, the stress tensor can be calculated directly from the strain tensor
FEM-TFM	Finite element method-based TFM	FEM can be used for iTFM in particular for large strains or complicated geometries
GF-TFM	Green's function-based TFM	For small strains and simple geometries, Green's functions can be used for iTFM
BEM	Boundary element method	iTFM in real space
FTTC	Fourier transform traction cytometry	iTFM in Fourier space
Reg-FTTC	Regularized FTTC	FTTC with regularization to deal with noise
HR-TFM	High resolution TFM	Reg-FTTC with differently colored marker beads
TRPF	Traction reconstruction with point forces	Reconstruction of point-like forces rather than stresses
MB-TFM	Model-based TFM	To avoid regularization and to infer forces beyond traction forces, models can complement TFM

In components this equation reads

$$\varepsilon_{ij} = \frac{1}{2} \left(\frac{\partial u_i}{\partial x_j} + \frac{\partial u_j}{\partial x_i} \right). \tag{6}$$

The linear strain tensor is non-dimensional and describes the relative length changes in different directions. For simplicity, for the following we assume small strains and thus proceed with the linear strain tensor (*geometrical linearity*); later we will come back to the large deformation case.

If in addition to small strains one can also assume that the substrate material has a linear and isotropic constitutive relation (*material linearity*), then a linear relation exists between the strain tensor ε and the stress tensor σ describing the forces acting over internal surfaces:

$$\sigma_{ij} = \frac{E}{1+\nu} \left(\varepsilon_{ij} + \frac{\nu}{1-2\nu} \varepsilon_{ll} \delta_{ij} \right) \tag{7}$$

where summation over repeated indices is implied. This equation can be inverted to

$$\varepsilon_{ij} = \frac{1}{E} \left[(1+\nu)\sigma_{ij} - \nu\sigma_{ll}\delta_{ij} \right]. \tag{8}$$

Here *E* and *v* represent the two elastic constants of the linear and isotropic substrate. For experiments with tissue cells like fibroblasts, one typically uses a Young's modulus $E \approx 10$ kPa; for weaker cells like neurons, one typically uses $E \approx 100$ Pa. The Poisson's ratio $v \approx 0.45$ for PAA-films [67] and $v \approx 0.5$ for PDMS-films [53]. Thus both material systems are close to being incompressible ($v \approx 0.5$).

The balance of internal and body forces, $\nabla \sigma = \mathbf{f}$, can be written as partial differential equation for the displacement vector field, the Lamé equation:

$$\frac{E}{2(1+\nu)}\Delta \mathbf{u} + \frac{E}{2(1+\nu)(1-2\nu)}\nabla(\nabla \cdot \mathbf{u}) = \mathbf{f}.$$
(9)

Due to its linearity, the general solution of Eq. (9) can be constructed from the Green's function, which is the solution for the external force **f** being a point force (represented mathematically by the delta function). For example, a point force solution for infinite elastic space can be calculated (*Kelvin solution*) [65]. This solution decays like 1/(Er) with Young's modulus and distance r (long-ranged decay) as the Lamé equation Eq. (9) is similar to a Laplace equation. The angular parts of the Green's function are determined mainly by the Poisson's ratio v.

In TFM, one typically has $\mathbf{f} = \mathbf{0}$ for the body forces and deformations arise from traction forces on the boundaries. However, due to the linearity of Eq. (9), a propagator can be calculated for point traction forces that in the following we also denote as Green's function. Most importantly for TFM, such Green's functions are known for the elastic halfspace (*Boussinesq solution*) [65] and for the elastic layer of finite thickness [55,59–61]. Once such a Green's function is known, the general solution follows as convolution integral:

$$\mathbf{u}(\mathbf{x}) = \int \mathbf{G}(\mathbf{x}, \mathbf{x}') \mathbf{t}(\mathbf{x}') d\mathbf{x}'. \tag{10}$$

The typical film thickness used in cell experiments is between $50 \,\mu m$ and $80 \,\mu m$. Then the displacement field decays sufficiently fast into the substrate such that the boundary conditions at the bottom do not matter and the Boussinesq solution can be used. For weak cell types, even 20 μm may suffice. For strong cell types and if a thick substrate does not fit into the microscope, one can use the Green's function for finite substrate thickness.

The basic equations of elasticity theory reviewed here now suggest two ways to construct forces from displacements. First one can invert the convolution integral from Eq. (10) as one wants to infer the traction forces **t** from the displacement data **u** (inverse TFM). Alternatively one can directly construct the strain tensor ε from the displacement field **u** using Eq. (6) and from this the stress tensor σ using Eq. (7). The traction forces then follow as **t** = σ **n** where **n** is the normal to the substrate surface (direct TFM). In the large deformation case, the linear relation from Eq. (7) has to be replaced by a non-linear material law. A standard choice is the Neo-Hookean model leading to [64]

$$\sigma = \frac{E}{2(1+\nu)\det(\mathbf{F})^{5/3}} \left(\mathbf{B} - \frac{1}{3}tr(\mathbf{B})\mathbf{1} \right) + \frac{E}{3(1-2\nu)} (\det(\mathbf{F}) - 1)\mathbf{1}$$
(11)

where **B** is the left Cauchy–Green tensor defined in Eq. (3). For small strains, this simplifies to the linear relation from Eq. (7). We note that Eq. (11) describes compressible materials and that the second term diverges in the incompressible limit v = 0.5. For this case multi-field formulations can be employed where incompressibility is treated separately [66].

It is essential to note that all TFM-methods suffer from the presence of noise in the experimental data. Noise in TFM-experiments has different origins, including the limited optical resolution of the microscope, the uncertainties in the image processing procedures and heterogeneities in the substrate material with its embedded marker beads. In the case of inverse TFM, the inversion of the convolution integral (a Fredholm integral equation of the 1st kind) in the presence of noise is ill-posed due to the long-ranged (1/r) nature of the elastic Green's function. This means that small changes in the displacement data can lead to large changes in the reconstructed forces, thus rendering the inverse procedure unstable. In the case of direct TFM, calculation of the strain tensor amounts to taking 3D derivatives of noisy data and therefore can be unstable, too. Therefore very good 3D image quality has to be achieved and numerical derivatives have to be calculated with appropriate noise filters, e.g. the optimal-11 differentiation kernel [25]. For standard data sets it is therefore more feasible to proceed with inverse TFM. Only recently has it become possible to implement direct TFM based on much improved image processing procedures. Therefore in the following we first discuss inverse TFM, in particular its main variants, the Fourier method FTTC, which is a special version of GF-TFM.

3. Green's function-based TFM and FTTC

The first quantitative reconstruction method to measure traction patterns with sub-cellular resolution was proposed for pre-stressed silicone sheets [17]. One disadvantage of this approach is that twodimensional Green's functions scale as ln(r). Because of the divergence with large r, boundary effects become even more important than in the case of the long-ranged but decaying Green's function with 1/r-scaling in three dimensions. However, single cells are isolated systems that cannot generate momentum and therefore to a first order approximation their force monopole vanishes and they act as force dipoles [68]. This means that the actual decay in such experiments should scale as 1/rrather than as ln(r) even in two dimensions. Nevertheless the twodimensional decay of the Green's function is slow and therefore problematic. For this reason this technique was adapted to the case of thick substrates [18], when the Green's function decays as $1/r^2$ for the force dipole. With this faster decay, the inversion procedure becomes more stable.

Due to the tensor nature of elasticity theory the Green's function **G** (either for a halfspace or a finite thickness film) is actually a 3×3 matrix, which accounts for 3D substrate displacements. In most cases it is sufficient and much more simple to record substrate deformations only in lateral directions (2D), however, in this case it is a mandatory requirement that the substrate material is nearly incompressible, which is fulfilled by a Poisson's ratio of $v \approx 0.5$. Only under these circumstances

the matrix entries in lateral and vertical directions decouple. We then have $\mathbf{G}(\mathbf{x},\mathbf{x}')=\mathbf{G}(\mathbf{x}-\mathbf{x}')$ with

$$\mathbf{G}(\mathbf{x}) = \frac{(1+\nu)}{\pi E r^3} \begin{pmatrix} (1-\nu)r^2 + \nu x^2 & \nu xy \\ \nu xy & (1-\nu)r^2 + \nu y^2 \end{pmatrix}.$$
 (12)

Here $r = \sqrt{x^2 + y^2}$ is the 2D distance. Note that **G**(**x**) can be inverted everywhere (*detG* > 0) except at the singularity at **x** = **0**. This singularity however is effectively removed by the integration procedure in the convolution integral from Eq. (10).

The inverse problem can be solved with Green's functions either in real space or in Fourier space. In real space, this is commonly done using the boundary element method (BEM) [17,18,21]. The boundary element method needs in addition to measured substrate deformations the shape of the cell area, which gets spatially discretized by triangulation. The traction field gets thereby represented by an interpolation scheme based on a finite set of nodal traction values (similar to the Galerkin method in FEM). Then Eq. (10) can be written as a linear algebraic equation, which represents a standard boundary element method. To invert the equation a priori information about expected traction field needs to be incorporated. This is usually implemented by introducing Tikhonov regularization. Here, an additional regularization term is added to the corresponding optimization problem that replaces the inversion of the elastic equation. This regularization term is weighted by a regularization parameter λ that determines the strength of regularization. In principle there are various choices of regularization terms possible which are characterized by their property to enhance or suppress specific traction distributions. Because cells on flat substrates may exert forces in different directions at neighboring adhesions, in contrast to many other application areas for regularization methods smoothness of the traction field (first-order Tikhonov regularization) is not an appropriate choice. In practice one finds that inversion without regularization can lead to very large forces as the algorithm tries to reflect small details of the noisy displacement field with large forces working against each other. The most reasonable assumption therefore is that traction forces should not become exceedingly large. One thus arrives at a minimization problem with zero-order Tikhonov regularization:

$$\mathbf{t} = min_{\mathbf{t}} \left((\mathbf{G}\mathbf{t} - \mathbf{u})^2 + \lambda^2 \mathbf{t}^2 \right)$$
(13)

where **u** is the experimentally measured displacement and **Gt** is the displacement predicted by the estimated traction (the convolution integral is not written out here because in practice it is discretized and thus becomes a matrix operation). The choice of the regularization parameter λ can be based on Bayesian theory [22] or determined empirically, for example with the L-curve criterion [69]. The BEM is a conceptually straightforward but computationally expensive method to reconstruct traction patterns. Its main advantages are that it can reach a high spatial resolution and that it constrains traction to the segmented cell area.

Alternatively one can work in Fourier space (Fourier transform traction cytometry, FTTC) [19]. In Fourier space the convolution integral Eq. (10) factorizes into a product and then can be inverted to give

$$\tilde{\mathbf{t}}(\mathbf{k}) = \mathbf{G}(\mathbf{k})^{-1} \tilde{\mathbf{u}}(\mathbf{k}). \tag{14}$$

Here the tilde denotes a Fourier transform. The traction reconstruction is achieved in principle by fast Fourier transform of the displacement field, multiplication with the inverse matrix and fast Fourier transform of the result back into real space. The 2D Green's function from Eq. (12) becomes in Fourier space

$$\mathbf{G}(\mathbf{k}) = \frac{2(1+\nu)}{Ek^3} \begin{pmatrix} (1-\nu)k^2 + \nu k_y^2 & -\nu k_x k_y \\ -\nu k_x k_y & (1-\nu)k^2 + \nu k_x^2 \end{pmatrix}.$$
 (15)

FTTC as suggested initially is not regularized and thus could lead to problems in the presence of noise. In practice, this issue can be attenuated by smoothing the image data and cutting away fast Fourier modes. A more rigorous approach however is to introduce a regularization scheme into FTTC (Reg-TFM) [21,22]. Eq. (14) is then replaced by

$$\tilde{\mathbf{t}} = \left(\mathbf{G}^{T}\mathbf{G} + \lambda^{2}\mathbf{1}\right)^{-1}\mathbf{G}^{T}\tilde{\mathbf{u}}$$
(16)

where we have left out the **k**-dependence for notational simplicity. For regularization parameter $\lambda = 0$, we get back the unregularized solution Eq. (14).

FTTC benefits from a very fast computation time and that no additional information apart from the measured displacement field is required. Currently it is the most widely used traction reconstruction technique and we show examples of FTTC in action in Figs. 2 and 3. FTTC has been implemented by many labs and also is available as ImageJ plugin [71]. When working with it, however, one should keep some caveats in mind. In general one has to note that the resolution of the final traction field strongly depends on the local resolution of substrate deformations according to the Saint-Vernant's principle, which states that the difference between the displacement fields caused by traction distributions that are different but have the same highest order force multipolar moment decays quickly with the distance from the traction source. Therefore in experiments it is essential to record the substrate deformation close to the force generating processes to resolve the details of the traction pattern. In practice this means that the deformations right under the cell body are most relevant. We next note that according to the Nyquist criterion, the spatial resolution is mainly determined by the bead density as it can be resolved in the image processing procedures. Therefore nanobeads are preferred over microbeads, which have the additional advantage of less disturbing the mechanical properties of the substrate. We also note that fast Fourier transform requires interpolation onto a regular lattice and therefore a homogeneous distribution of marker beads is essential. We further note that for high resolution TFM, one should also keep track of the depth at which the marker bead images are taken; then an appropriately modified Green's function in Fourier space has to be used [22,55]. In general, these comments again show that the force reconstruction cannot be separated from the quality of the image data and its processing.

4. FEM-based TFM

Standard TFM assumes both geometrical and material linearity. If these assumptions are not justified anymore, then finite element methods (FEM) can be used. This might be the case for soft substrates or strong cells, and usually is the case if cells are embedded in a 3D elastic matrix [63,72–74]. Moreover FEM-TFM can also be used in the linear case [62,63]. Here we briefly comment on some applications of this approach.

We first note that the boundary element method (BEM) introduced early by Dembo and Wang [18] can be considered to be some variant of FEM-TFM. However, here only the cell and not the substrate is discretized. It has been shown that TFM can also be completely based on such a discretized approach, for example in order to take into account finite substrate thickness [62]. Here the agreement between measured and simulated displacements has been optimized using a FEM-formulation. A mathematically very elegant way to reconstruct traction with FEM has been introduced by Ambrosi and coworkers [75,76]. The basic idea here is to consider the traction reconstruction as PDE constraint optimization problem. Thus one searches for the stationary state of a Lagrangian type functional that incorporates the elastic PDE as Lagrangian restriction. This leads to a coupled system of Lamétype equations associated with the forward and the adjoint problems. By doing this one prevents costly numeric optimization procedures since the optimization problem is reduced to the solution of a coupled

system of PDEs, which can be solved in parallel. One disadvantage of this approach is that it assumes thin substrates and plane stress.

For thick substrates, another FEM-approach has been suggested by Hur and coworkers [56,77]. They avoid the inversion of the elastic equations by considering the elastic problem as mixed boundary value problem (BVP). Here, FEM is used as suitable tool to solve the elastic equations with respect to the Cauchy stress σ , while measured bead displacements enter the equations as boundary condition. The substrate region is considered as cuboid discretized domain and the equation system that has to be solved is represented by inversion of the constitutive relation from Eq. (8). The cellular traction at the substrate surface follows as $\mathbf{t} = \sigma \mathbf{n}$. The approach does not use regularization and therefore might be vulnerable to noise. The authors validated their method with simulated data contaminated with additive Gaussian noise of different strengths and measured a relative deviation of the root mean square $\Delta t = ||t_{sim} - t_{recon}||$ of traction up to 25% for a Gaussian noise standard deviation of 200 nm. They further applied the method to data sets of bovine aortic endothelial cells (BAECs) and found prominent non-zero traction stresses in normal direction. These normal forces have now been measured also in other approaches that reconstruct z-direction forces [24,57,58].

Another FEM-based approach was proposed by Legant and coworkers to reconstruct traction force patterns of cells embedded in 3D elastic matrix [63]. Compared to the BVP approach by Hur and coworkers, this method utilizes FEM not to solve the problem directly, but to calculate a discretized Green's function based on each individual data set including the segmented and triangulated 3D cell shape. For each facet of the cell surface triangulation the displacement responses for unit loads are calculated with FEM to construct an individual discrete Green's function. Based on the calculated Green's function then traditional traction reconstruction is applied using inversion of the convolution formulation from Eq. (10). Legant et al. inverted the problem by minimizing a least square estimate with a zero-order Tikhonov with respect to the facet loads similar to the TRPF method. The described method represents a smart hybrid method. While it uses the strength of FEM numerics to calculate PDEs on complex geometries, it also traces the problem back to well-established standard TFM. Compared to the Hur approach this represents a higher degree of flexibility, e.g. beads do not have to be distributed directly at the substrate surface. Because this is a Green's function approach, however, this method is limited to linear elasticity theory after all.

Fig. 2 depicts a direct comparison of traction reconstruction with standard FTTC and a FEM-based TFM for a cardiac myocyte on a E = 15 kPa PDMS substrate. The implementation of the FEM-based method used here is described in detail in [70]. The general idea behind this technique is the replacement of the convolution integral Eq. (10) by the direct elastic PDE, which is solved with respect to a freely chosen traction boundary condition. This boundary condition is parameterized by an interpolation scheme, which consists of a certain number of traction degrees of freedom analogue to the discretization used in FTTC. Traction fields here are reconstructed by repeated calculation of the PDE, while optimizing the traction boundary condition with respect to measured bead displacement fields (beads distributed either on substrate surface or within substrate volume). Note that by replacing the convolution formulation (Eq. (10)) by the FEM-calculation step all constrictions regarding material model and substrate geometry can be overcome, which designates the method to applications like nonplanar geometries and non-linear substrate material behavior. Visual comparison of Fig. 2 c and d demonstrates that both methods lead to identical traction fields and resolutions.



Fig. 2. Example of force reconstruction with different methods [70]. (a) Phase contrast image of a cardiac myofibroblast on an *E* = 15 kPa PDMS-substrate. (b) Substrate displacement field derived from the displacement of embedded fluorescent marker beads. (c) Reg-FTTC traction reconstruction. (d) FEM-based traction reconstruction. Data courtesy of Merkel group.



Fig. 3. Model-based traction force microscopy (MB-TFM) with active cable networks [52]. U2OS-cell on E = 8.4 kPa PAA substrate. (a) Actin fluorescence image. Scale bar is 10 μ m. (b) Paxillin fluorescence image. (c) Segmented actin stress fibers from actin image. (d) Segmented focal adhesions from paxillin image. (e) Reg-FTTC traction force pattern from substrate deformations. (f) MB-TFM traction force pattern based on an active cable cell model. The insets illustrate the different levels of detail obtained from FTTC versus MB-TFM reconstructions. Data courtesy of Gardel group.

5. Model-based traction force microscopy

All previous TFM reconstruction methods focused on finding a correct distribution of traction forces with respect to a measured field of substrate deformations. However, this does not tell us how forces are distributed inside the cells. This restriction has been lifted for cell monolayers by the introduction of monolayer stress microscopy which models the cell monolayer as thin elastic sheet [42,43,50]. The strategy of combining traction reconstruction with modeling assumptions is very general and recently has also been applied to the stress fibers in single cells as an example of model-based TFM [52].

The main idea behind MB-TFM is to combine TFM with quantitative cell modeling. In practice image processing is used to identify prominent mechanical features of the cell based on fluorescence images of focal adhesion proteins and the actin cytoskeleton. Fig. 3 a-d shows how these structures can be segmented in live cells (TFM microscopy is only possible in combination with live cell microscopy due to the need to also acquire a reference image). This information is subsequently used to build an individual 2D biophysical model for each cell that represents their mechanical behavior as best as possible while keeping a moderate level of model parameters. For this purpose the cell is represented by a network of contractile active cables. Segmented stress fibers are embedded in the network and after its simulated contraction the resulting forces are transmitted to the substrate at those points where focal adhesion have been segmented before. By doing so, one arrives at a quantitative mechanical cell model with typically 30 to 100 free parameters to optimize, which is much less compared to traditional TFM reconstruction. A comparison of both reconstructions is shown in Fig. 3 e and f.

MB-TFM has several advantages compared to standard techniques. First, the incorporation of the cell model allows us to reveal correlations between substrate deformations and the distribution of intracellular tension. E.g. for U2OS cells it could be shown that forces generated in actin stress fibers make up most of the overall traction force, while forces originated from distributed actin networks are in most of the cases negligible. Further it could be validated that different stress fiber types, which have been classified by their assembly mechanism, vary statistically in their contraction strength. Second, the additional model restriction and according reduction of the parameter space is sufficient to abolish the need for regularization in most cases, while the model filter undertakes the task of a regularization scheme and guarantees that traction field solutions agree with biophysical considerations including e.g. the overall balance of forces. The drawback of this approach is the restriction to a 2D model, the need for very good image data (in order to perform the segmentation and to arrive at a realistic model), the requirement to image different channels simultaneously (one for the marker beads and at least two for adhesion and cytoskeletal proteins, respectively) and the high demand on computer power. With the rapid advances in 3D imaging, in the future this approach might also be carried into the third dimension. However, although 3D models for contractility of adherent cells already exist [78,79], it is certainly challenging to acquire the image data needed to parameterize such 3D models through traction data.

6. Experimental data and image processing

We finally comment on some practical issues that arise in TFM. A typical TFM-experiment proceeds as follows. Polymer monomers and crosslinkers are mixed at a ratio that corresponds to the desired substrate stiffness. Then marker beads (commonly used diameters range from µm down to 40 nm) are added to this mixture and a drop is deposited on a first coverslip. A second coverslip is used to sandwich the drop and the gel is polymerized. The top coverslip is removed again and the surface coated with extracellular matrix. This culture substrate is then inserted into the microscope. Sometimes the beads are centrifuged down towards the substrate surface in order to facilitate image processing. It is difficult to dissolve the beads in the substrate material (e.g. in PDMS), then the substrate can be made in a two-step process with the beads deposited on the surface of the first film. The rigidity of the final substrate can be measured with different techniques, ranging from microscopic probes like the AFM or magnetic beads to macroscopic probes like film stretching under gravity or rheometers. Ideally measurements with different setups should give the same results.

Over the years, different protocols have been published for the preparation of TFM-substrates. An early and very rich resource is volume 83 of Methods in Cell Biology edited by Yu-Li Wang and Dennis E. Discher [54]. The standard system used for TFM are polyacrylamide (PAA) substrates that can easily be tuned over a large stiffness range [67,80,81]. For high resolution TFM, these protocols can be modified to use differently colored marker beads [22]. Surface functionalization is usually achieved by activating the surface with Sulfo-SANPAH or hydrazine, which then can bind ECM-ligands such as collagen or fibronectin. Because acrylamide monomers are toxic (for example, they react with the amino acid cysteine), care has to be taken that the substrates are indeed completely polymerized before being used for cell culture. Typically one prepares a stock solution that can be kept for several years and then prepares working solution on demand.

An alternative system to PAA-substrates is silicone-oil based substrates (mainly PDMS, that is commercially available as Sylgard 184) for which also different protocols are available [53,55]. Their high index of refraction makes them suitable for simultaneous use with TIRF [82]. However, they are also more difficult to handle in regard to inclusion of marker beads, which often are added in a multiple-step process between two PDMS-layers [55]. Alternatively, the good patterning properties of PDMS can be used to create some surface features that can be followed in phase contrast [3]. Another issue with PDMS-substrates is that they are very sticky and therefore hard to handle; at the same time, it is difficult to achieve the kPa-stiffness needed for cell experiments. Motivated by the widespread use of TFM-substrates in mechanobiology, recently the difference between PAA- and PDMS-substrates for stem cell differentiation have been investigated in great detail [14,15].

An exciting development in TFM is the combination with micropatterning approaches, which allow one to normalize cell shape and intracellular organization [83]. Different techniques have been developed to pattern soft elastic substrates, including microcontact printing [84–87] and deep-ultraviolet illumination of PAA-substrates [88–90] as well as lift-off techniques on PDMS-substrates [91]. Combining TFM and cell patterning can also be accomplished in three dimensions, e.g. using direct laser writing [92,93].

When working with soft elastic substrates, one always has to keep in mind that many different physical processes might cause them to deviate from the ideal elastic behavior assumed in the force reconstruction procedure. For the soft kPa-substrates required for TFM, care has to be taken to avoid viscous components (e.g. PDMS at 0.1 kPa [14]) or heterogeneities (e.g. PAA at unfavorable mixing ratios [80]) because both cell behavior and the force reconstruction procedures are affected by such substrate properties. If substrates with dissipative elements are used, their viscoelastic properties (creep and relaxation functions) should be characterized in detail [94]. For the widely used PAA-substrates, another effect that one has to keep in mind is the possible uptake and release of water by the deformed hydrogel [95].

In a TFM experiment, one typically acquires a stack of fluorescence images, including at least one image for the substrate deformation. At the end of the experiment, one has to remove the cell (typically by trypsination or by adding similar enzymes, but alternatively by scratching or by adding calyculin such that the cell rips itself off the substrate) in order to obtain a reference image. Then the deformation vector field has to be constructed by image processing and comparison of the deformed and the undeformed images. There exist two standard solutions plus a number of hybrid approaches [55]. In single particle tracking, one uses a low number of fiducial markers such that one can track them one by one. In correlative tracking, also known as particle image velocimetry (PIV) or digital image correlation (DIC), one tracks patterns rather than single particles. Often one uses a combination of both approaches, e.g. PIV to get a first estimate and then single particle tracking (possibly augmented by neighborhood features) for a more precise one. The 3D variant of DIC is known under the term digital volume correlation (DVC). In addition, in HR-TFM, one can use the correlation between different colors [21,22].

Over the last years, different software packages have been developed for the image processing and force reconstruction tasks. In Table 2 we list some of them for the convenience of the TFM-user. The ImageJ plugin for FTTC is a ready-to-use solution to implement standard TFM [71]. The finite thickness software is required when cells are very strong or the substrate is very thin [55]. The regularization tools by Hansen can be used for regularized optimization strategies in inverse TFM based on optimization, compare Eq. (13) [69]. Helpful image processing tools are available also from other fields like environmental physics, fluids dynamics, microrheology, single molecule imaging or super resolution microscopy. A standard choice for correlative tracking is the MatLab PIV toolbox. Alternatively one can use single particle tracking routines. For large deformation 3D TFM, a specialized code has been developed very recently [25].

7. Conclusion and outlook

Here we have reviewed recent progress in TFM on soft elastic substrates from the computational perspective. Our overview shows that this field is moving at a very fast pace and that many different variants of this approach have been developed over the last two decades, each with its respective advantages and disadvantages. Thus the situation is similar to the one for optical microscopy, a field in which also a lot of progress has been made over the last decades at many different fronts simultaneously, ranging from super resolution microscopy (STED, PALM, STORM, SIM) through correlative and fluctuation microscopy to light sheet microscopy (SPIM). Like in this field, too, for TFM the choice of method depends on the experimental question one is addressing and in many cases a combination of different approaches will work best.

Our review shows that force reconstruction cannot be separated from data analysis and image processing, in particular due to the noise issue. Irrespective of the approach used, care has to be taken to deal with the experimental noise that is always present in the displacement data. Each of the methods discussed above includes some kind of regularization, either implicitly through image filtering or explicitly through some regularization scheme.

For 2D TFM, the standard approach is FTTC and this has become the common procedure in many labs working in mechanobiology due to its short computing times. Reg-FTTC makes computation time only slightly longer but introduces a more rigorous treatment of the noise issue. FTTC can be extended easily to 3D TFM, but only if the image data is of very good quality. The simplest version of 3D TFM implies tracking of marker beads in z-direction on planar substrates. As such experiments usually require relatively soft substrates, one typically leaves the linear domain and large deformation methods have to be used for force reconstruction. Similar techniques can then be used also in full 3D TFM, e.g. when cells are encapsulated in hydrogels. Here however care has to be taken that the cellular environment is indeed elastic; otherwise a viscoelastic or even plastic theory has to be employed.

The more complex the questions and experiments become that are conducted in mechanobiology, the more difficult it will get to extract meaningful correlations and cause–effects relations. We therefore envision that in the future, such experiments will be increasingly combined

Table 2

List of helpful software for TFM.

Description	URL
ImageJ plugin FTTC by Qingzong Tseng	https://sites.google.com/site/qingzongtseng/tfm
Finite thickness TFM from Dufresne group	http://dx.doi.org/10.1039/C4SM00264D
Regularization tools by Per Hansen	http://www.mathworks.com/matlabcentral/fileexchange/52-regtools
MatLab PIV toolbox by Nobuhito Mori	http://www.oceanwave.jp/softwares/mpiv/
Particle tracking code by Daniel Blair and Eric Dufresne	http://physics.georgetown.edu/matlab/
Large deformation 3D TFM from Franck group	http://franck.engin.brown.edu/~christianfranck/FranckLab/Downloads.html

with mathematical models that allow us to extract useful information from microscopy data in a quantitative manner. Simple examples discussed above are TRPF and MB-TFM, which use the assumptions of localization of force transmission to the adhesion contacts (TRPF) and force generation in the actin cytoskeleton (MB-TFM) to improve the quality of the data that one can extract from TFM-experiments. In a similar vein, we expect that in the future, more and more data will be extracted from microscopy images based on some Bayesian assumptions that have been validated before by other experimental results. Another very exciting development is the combination of TFM with fluorescent stress sensors, which complement it with molecular information and which can be more easily used in a tissue context.

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